

# Defect Engineering and 2D/2D Coupling in Mn- and B-Doped BiVO<sub>4</sub> and Ti<sub>3</sub>C<sub>2</sub> Nanostructures

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## ABSTRACT

Doping is an effective method that modulates the electronic and optical properties of semiconductor nanomaterials through the addition of foreign atoms. In this work, bismuth vanadate (BiVO<sub>4</sub>) and titanium carbide (Ti<sub>3</sub>C<sub>2</sub>) MXene nanosheets were synthesized through facile solution processes. To elucidate the roles of defect engineering and 2D/2D interfacial coupling, a series of systems including Mn-BiVO<sub>4</sub>, B-Ti<sub>3</sub>C<sub>2</sub>, Ti<sub>3</sub>C<sub>2</sub>/BiVO<sub>4</sub>, Mn-BiVO<sub>4</sub>/Ti<sub>3</sub>C<sub>2</sub>, BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub>, and Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub> were synthesized and characterized by XRD, FTIR, FESEM, and EDX. DRS results reveal that Mn and B doping, as well as the construction of the heterostructure, lower the optical band gap from 2.9 eV (pristine BiVO<sub>4</sub>/Ti<sub>3</sub>C<sub>2</sub>) to 2.4 eV (Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub>). These controlled defect states and 2D/2D interfaces enhance charge separation and photoactivity, demonstrating the potential of Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub> for photo-driven redox applications.

**KEYWORDS:** Ti<sub>3</sub>C<sub>2</sub> MXene, Doping, Heterostructures, Photoactivity

## 1. Introduction

MXenes represent a novel class of two-dimensional (2D) nanomaterials, composed of transition metal carbides, nitrides, or carbonitrides, with the general formula M<sub>n+1</sub>X<sub>n</sub>, where M denotes an early transition metal and X correspond to carbon and/or nitrogen [1,2]. Since their initial discovery in 2011, the diversity of MXenes has significantly expanded, leading to extensive investigations into their physicochemical properties and potential applications. [3,4]. Titanium carbide (Ti<sub>3</sub>C<sub>2</sub>) is one of the most popular among them due to its ability to attract water, electrical conductivity, high surface area, stability in chemicals, and biocompatibility. It also possesses a tunable electronic band structure. These characteristics render Ti<sub>3</sub>C<sub>2</sub> MXene suitable for applications in energy storage, catalysis, and electronics [5,6]. Ti<sub>3</sub>C<sub>2</sub> is typically prepared through the selective etching of Ti<sub>3</sub>AlC<sub>2</sub> MAX phase precursors. This process results in 2D

nanosheets bearing numerous surface groups (-OH, -O, -F) that facilitate their ability to disperse in water and strengthen their interaction with other nanomaterials [7-9].

Bismuth vanadate (BiVO<sub>4</sub>) is an n-type semiconductor with a band gap of ~2.4 eV, which is stable, inexpensive, and environmentally friendly, making it a promising material for photoelectrochemical (PEC) water splitting [10-12]. Pristine BiVO<sub>4</sub>, however, is plagued by inefficient charge transport and fast electron-hole recombination, resulting in low photocatalytic efficiency [13-15]. To overcome these limitations, various strategies have been suggested, including foreign element doping and coupling with highly conductive 2D nanostructures. Doping can tailor the electronic structure through the introduction of defect states, which can improve carrier diffusion, conductivity, and light harvesting [16-22]. Substitutional metal dopants, in particular,



have been found to enhance photoactivity through the elongation of carrier lifetimes and adjustment of band positions [23–25]. Hybridization with conductive nanosheets like  $\text{Ti}_3\text{C}_2$  MXene, on the other hand, promotes interfacial charge separation and directional electron transport, providing another avenue for performance improvement [26–30]. In this research, we synthesized a series of nanostructures comprising pure  $\text{Ti}_3\text{C}_2$  and  $\text{BiVO}_4$ , Mn-doped  $\text{BiVO}_4$  (Mn- $\text{BiVO}_4$ ), B-doped  $\text{Ti}_3\text{C}_2$  (B- $\text{Ti}_3\text{C}_2$ ), and their hybrids ( $\text{Ti}_3\text{C}_2/\text{BiVO}_4$ , Mn- $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ ,  $\text{BiVO}_4/\text{B-Ti}_3\text{C}_2$ , and Mn- $\text{BiVO}_4/\text{B-Ti}_3\text{C}_2$ ) via solution-based processes. We conducted examinations on their structure, morphology, and optical properties. Therefore, our work reports a dual-defect/dual-2D coupling design strategy that systematically demonstrates how Mn and B dopants, coupled with  $\text{Ti}_3\text{C}_2$ - $\text{BiVO}_4$  nanosheet coupling, cooperatively modulate the structure property relationship.

## 2. Experimental

### 2.1. Materials and apparatus

The  $\text{Ti}_3\text{AlC}_2$  MAX phase precursor was procured from Redox Kala Co. (Iran). All other chemicals used in this study were of analytical grade and obtained from Merck, including hydrofluoric acid (HF, 40%), sodium metavanadate ( $\text{NaVO}_3$ ,  $\geq 99\%$ ), bismuth(III) salt ( $\text{Bi}_5\text{H}_9\text{N}_4\text{O}_{22}$ ), cetyltrimethylammonium bromide (CTAB,  $\geq 99\%$ ), boric acid ( $\text{H}_3\text{BO}_3$ ,  $\geq 99.5\%$ ), manganese(II) nitrate tetrahydrate ( $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$ ,  $\geq 98\%$ ), ethylene glycol ( $\text{C}_2\text{H}_6\text{O}_2$ ,  $\geq 99\%$ ), ethanol ( $\text{C}_2\text{H}_5\text{OH}$ , 96%), sulfuric acid ( $\text{H}_2\text{SO}_4$ , 98%), and ammonium metavanadate ( $\text{NH}_4\text{VO}_3$ ,  $\geq 99\%$ ). Deionized water (DIW) was used throughout the synthesis processes to ensure the purity of the nanostructures. The synthesized nanomaterials were characterized using various advanced analytical techniques. Fourier-transform infrared (FTIR) spectra were recorded on a Bruker Tensor 27 spectrometer to identify the functional groups and confirm the chemical composition. The crystallographic structure and phase purity of the synthesized samples were analyzed using an X-ray diffractometer (Rigaku D-max C III, Cu K $\alpha$  radiation,  $\lambda = 1.5406 \text{ \AA}$ ). The surface morphology and elemental composition were examined via field-emission scanning electron microscopy (FESEM) and energy-dispersive X-ray spectroscopy (EDX) using a MIRA3 TESCAN-XMU instrument. The optical properties and electronic band structure

modifications were investigated through diffuse reflectance spectroscopy (DRS) using a Shimadzu UV-160A spectrophotometer.

### 2.2. Synthesis of $\text{Ti}_3\text{C}_2$ and B- $\text{Ti}_3\text{C}_2$

$\text{Ti}_3\text{C}_2$  MXene nanosheets were synthesized through a modified selective etching process. 1 g of  $\text{Ti}_3\text{AlC}_2$  MAX phase powder was added slowly into 10 mL of 40% hydrofluoric acid (HF) under continuous stirring at 25 °C for 24 h. The resulting dark suspension was centrifuged, and the precipitate was washed several times with deionized water (DIW) until the supernatant pH was close to 6.0. The thus purified  $\text{Ti}_3\text{C}_2$  was dried in a vacuum oven at 60 °C for 12 h. For exfoliation, the powder of  $\text{Ti}_3\text{C}_2$  was dispersed in 40 mL of the binary solvent system dimethyl sulfoxide (DMSO) and ethanol (1:1 v/v). The dispersion was stirred at room temperature for 1.5 h, followed by probe sonication (400 W, 7 s on/3 s off) for 30 min to achieve delamination of the layered structure of  $\text{Ti}_3\text{C}_2$ . The resulting suspension was centrifuged at 5000 rpm for 10 min, washed repeatedly with DIW to remove the residual solvents, and finally centrifuged at low speed (1000 rpm, 10 min) to collect the supernatant containing the exfoliated  $\text{Ti}_3\text{C}_2$  nanosheets with colloidal stability in solution.

Boron-doped  $\text{Ti}_3\text{C}_2$  (B- $\text{Ti}_3\text{C}_2$ ) nanostructures were synthesized via a hydrothermal doping process. Typically, 0.1 g of  $\text{Ti}_3\text{C}_2$  MXene was suspended in 25 mL of 0.1 M aqueous boric acid ( $\text{H}_3\text{BO}_3$ ) solution and ultrasonicated (40 kHz, 20 min) for uniform dispersion. The suspension was poured into a Teflon-lined stainless-steel autoclave and maintained at 200 °C for 5 h under autogenous pressure. After the cooling to room temperature naturally, the B- $\text{Ti}_3\text{C}_2$  MXene was collected, washed with DIW, and dried under vacuum at 80 °C. The concentration of boric acid precursor was carefully controlled to maintain consistent doping levels. The synthesis was repeated, yielding samples with reproducible XRD and FTIR features.

### 2.3. Synthesis of $\text{BiVO}_4$ and Mn- $\text{BiVO}_4$

$\text{BiVO}_4$  nanosheets were synthesized by a hydrothermal route, as reported in our previous work [31]. Manganese-doped  $\text{BiVO}_4$  (Mn- $\text{BiVO}_4$ ) was prepared by doping  $\text{Mn}^{2+}$  into the  $\text{BiVO}_4$  lattice through a controlled doping procedure. In a standard synthesis, 0.02 g of  $\text{Mn}(\text{NO}_3)_2 \cdot 4\text{H}_2\text{O}$  (equivalent to  $\sim 2 \text{ mol\%}$  relative to  $\text{Bi}^{3+}$ ) was dissolved in 10 mL of deionized water and dripped

into 50 mL of ethylene glycol solution with 4 mL of  $\text{Bi}_5\text{H}_9\text{N}_4\text{O}_{22}$  solution under agitation. In order to ensure uniform doping, 50 mL of aqueous ammonium metavanadate ( $\text{NH}_4\text{VO}_3$ ) solution was added slowly over 30 min and stirred for 4 h at 25 °C. The suspension was then transferred to a Teflon-lined autoclave and treated hydrothermally at 120 °C for 36 h. It was cooled to room temperature, and Mn– $\text{BiVO}_4$  nanosheets were obtained, washed with DIW, and dried at 60 °C. Control of doping levels was achieved by precisely adjusting the Mn precursor concentration, while all other parameters (temperature, reaction time, solvent composition) were kept constant. Repeated syntheses confirmed reproducibility, with consistent XRD and FTIR patterns obtained across all batches.

#### 2.4. Synthesis of Mn– $\text{BiVO}_4$ /B– $\text{Ti}_3\text{C}_2$

The hybrid nanostructure Mn– $\text{BiVO}_4$ /B– $\text{Ti}_3\text{C}_2$  was fabricated by a solution-phase self-assembly approach assisted by ultrasonic dispersion and magnetic stirring. In general, 15 mg of Mn– $\text{BiVO}_4$  and 15 mg of B– $\text{Ti}_3\text{C}_2$  were co-dispersed into a mixed solvent of 100 mL ( $\text{H}_2\text{O}/\text{EtOH}$ , v/v = 1:1) to enhance colloidal stability and ensure uniform hybridization. The suspension was sonicated (40 kHz, 200 W, 30 min) at 25 °C room temperature to achieve effective exfoliation and homogenization of the nanosheets. The blend was subsequently stirred magnetically for 1.5 h to optimize interfacial coupling of Mn– $\text{BiVO}_4$  and B– $\text{Ti}_3\text{C}_2$ . The resulting hybrid was subsequently filtered under vacuum and vacuum-dried at 80 °C for 24 h in an attempt to eliminate the residual solvents and stabilize the

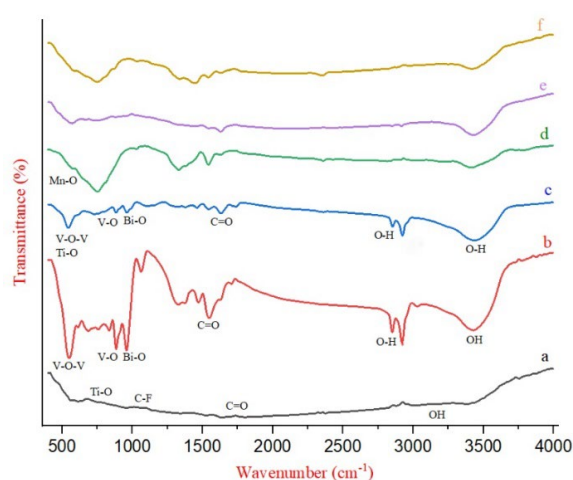
nanocomposite. Control of the hybrid composition was achieved by maintaining a 1:1 weight ratio of Mn– $\text{BiVO}_4$  to B– $\text{Ti}_3\text{C}_2$  in all syntheses, while keeping solvent composition, sonication power, and reaction time constant. For comparison, other heterostructures including  $\text{BiVO}_4$ /B– $\text{Ti}_3\text{C}_2$ , Mn– $\text{BiVO}_4$ / $\text{Ti}_3\text{C}_2$ , and  $\text{Ti}_3\text{C}_2$ / $\text{BiVO}_4$  were fabricated by replacing the respective precursors by following the same protocol.

### 3. Results and Discussion

#### 3.1. Characterization of the samples

##### 3.1.1. The effect of doping and coupling on chemical bonds

Fourier-transform infrared (FTIR) spectroscopy was employed to investigate the chemical bonding characteristics and functional group modifications resulting from doping and coupling in the synthesized nanostructures. As depicted in Fig. 1a, the FTIR spectrum of pristine  $\text{Ti}_3\text{C}_2$  MXene exhibits characteristic peaks at 616.15  $\text{cm}^{-1}$ , 955.38  $\text{cm}^{-1}$ , and 1633.84  $\text{cm}^{-1}$ , corresponding to the Ti–O stretching, C–F stretching, and C=O stretching vibrations, respectively. Additionally, a broad absorption band within the range of 3000–3500  $\text{cm}^{-1}$  is attributed to the O–H stretching vibrations, indicative of surface-terminated hydroxyl groups, which enhance hydrophilicity and facilitate interaction with other nanomaterials. The FTIR spectrum of  $\text{BiVO}_4$  (Fig. 1b) was previously reported in our work [31], and is reproduced here for comparison with the newly synthesized doped samples. Peaks at 550  $\text{cm}^{-1}$ , 884  $\text{cm}^{-1}$ , and 960  $\text{cm}^{-1}$ , corresponding to V–O–V bridging



**Fig. 1.** FTIR spectra of (a)  $\text{Ti}_3\text{C}_2$ , (b)  $\text{BiVO}_4$ , (c)  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ , (d) Mn– $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ , (e)  $\text{BiVO}_4$ /B– $\text{Ti}_3\text{C}_2$ , (f) Mn– $\text{BiVO}_4$ /B– $\text{Ti}_3\text{C}_2$  of nanostructures.

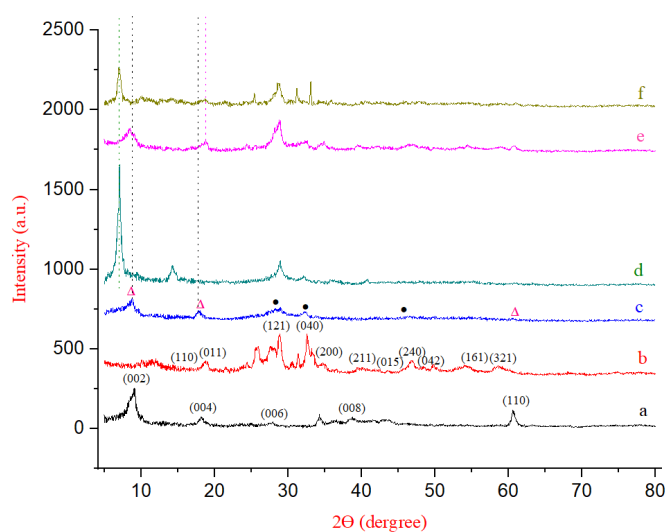
vibrations, V–O stretching, and Bi–O stretching modes, respectively. A peak observed at  $1548\text{ cm}^{-1}$  is attributed to the presence of carbonyl (C=O) groups, while additional absorption bands at  $2804\text{ cm}^{-1}$ ,  $2917\text{ cm}^{-1}$ , and  $3442\text{ cm}^{-1}$  correspond to O–H stretching, N–H bending, and hydroxyl (–OH) stretching vibrations, respectively. These functional groups play a crucial role in the interaction between  $\text{BiVO}_4$  and  $\text{Ti}_3\text{C}_2$  MXene during hybrid formation. Upon coupling of  $\text{BiVO}_4$  with  $\text{Ti}_3\text{C}_2$  and subsequent doping with Mn and B, significant spectral shifts and new vibrational modes emerge, confirming successful structural modification. As shown in Fig. 1c–1f, the presence of a distinct peak at  $560.1\text{ cm}^{-1}$  corresponds to Mn–O stretching vibrations, verifying the incorporation of Mn within the  $\text{BiVO}_4$  lattice in  $\text{Mn-BiVO}_4/\text{Ti}_3\text{C}_2$  and  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  nanostructures. The observed shifts in peak positions and intensities suggest changes in chemical bonding environments, likely due to electronic interactions and structural rearrangements induced by doping and coupling. These modifications are expected to influence the charge transfer dynamics and enhance the optical and electronic properties of the hybrid nanomaterials, making them promising candidates for advanced photocatalytic and optoelectronic applications.

### 3.1.2. The effect of doping and coupling on crystal-line structure

X-ray diffraction (XRD) analysis was conducted

to investigate the crystallographic phases and structural modifications induced by doping and coupling in the synthesized nanomaterials. As shown in Fig. 2a, the diffraction peaks at  $2\theta = 9.1^\circ$ ,  $18.15^\circ$ ,  $28.05^\circ$ ,  $38.79^\circ$ , and  $60.45^\circ$  correspond to the (002), (004), (006), (008), and (110) lattice planes of  $\text{Ti}_3\text{C}_2$  MXene, respectively, confirming its characteristic layered structure. For pristine  $\text{BiVO}_4$  (Fig. 2b), distinct diffraction peaks observed at  $2\theta = 17.28^\circ$ ,  $17.97^\circ$ ,  $29.03^\circ$ ,  $32.62^\circ$ ,  $35.52^\circ$ ,  $40.23^\circ$ ,  $43.54^\circ$ ,  $46.72^\circ$ ,  $48.11^\circ$ ,  $55.85^\circ$ , and  $58.75^\circ$  are indexed to the (110), (011), (121), (040), (200), (211), (015), (240), (042), (161), and (321) planes, respectively, which confirmed high purity of the monoclinic  $\text{BiVO}_4$  phase (JCPDS No. 14-0688).

The addition of  $\text{Ti}_3\text{C}_2$  to  $\text{BiVO}_4$ , as well as the addition of Mn and B dopants (including Fig. 2c–2f), gives rise to noticeable shifts in peaks as well as intensities. Specifically, the (110) peak of  $\text{BiVO}_4$  shifts downward from  $2\theta = 17.28^\circ$  to  $18.77^\circ$  after doping with Mn, corresponding to a decrease in interplanar spacing from  $5.13\text{ \AA}$  to  $4.72\text{ \AA}$ . This noticeable lattice contraction is consistent with the substitution of the larger-sized  $\text{Bi}^{3+}$  cations ( $1.17\text{ \AA}$ ) with the smaller-sized  $\text{Mn}^{2+}$  cations ( $0.80\text{ \AA}$ ). Similarly, the (002) peak of  $\text{Ti}_3\text{C}_2$  moved downward from  $2\theta = 9.1^\circ$  ( $d = 9.71\text{ \AA}$ ) to  $7.02^\circ$  ( $d = 12.58\text{ \AA}$ ) after being doped with B, indicating an increase of about  $2.87\text{ \AA}$  in interlayer separation, which may be attributed to the incorporation of boron species as well as their interactions with the surface-termination groups residing on  $\text{Ti}_3\text{C}_2$ .



**Fig. 2.** XRD patterns of (a)  $\text{Ti}_3\text{C}_2$ , (b)  $\text{BiVO}_4$ , (c)  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ , (d)  $\text{BiVO}_4/\text{B-Ti}_3\text{C}_2$ , (e)  $\text{Mn-BiVO}_4/\text{Ti}_3\text{C}_2$ , (f)  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  nanostructures

A comparative analysis of Fig. 2c ( $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ ) and Fig. 2f ( $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$ ) supports the successful construction of the hybrid structure, as evidenced by the appearance of the  $\text{Ti}_3\text{C}_2$  and  $\text{BiVO}_4$  peaks, together with corresponding shifts. The results present that the doping of Mn and B introduces noticeable lattice deformities, as expected to influence considerably charge transport as well as photocatalytic activity.

### 3.1.3. The effect of doping and coupling on size and morphology

Field-emission scanning electron microscopy (FESEM) was also used to study the dimensional changes and morphological features caused by doping and coupling of the nanostructures that are being synthesized. Fig. 3 shows FESEM images of the respective composites of the  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$  and  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  composites. From Fig. 3(a) and 3(b), it is portrayed that the  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$  composite has a heterogeneous sheet-like structure, wherein ultrathin nano-sheets of  $\text{BiVO}_4$  are dispersed on the broader  $\text{Ti}_3\text{C}_2$  MXene sheets. The nano-sheets of  $\text{BiVO}_4$  are of an average thickness of around  $10 \pm 2$  nm and of varying lateral dimensions of 50 to 150 nm, supporting the nature of nanosheet growth. Irregular distribution of  $\text{BiVO}_4$  reveals partial

coverage, thus likely affecting interfacial charge transfer nature.

After doping with Mn (Fig. 3(c, d)), there is a sudden increase in the lateral dimension of the  $\text{BiVO}_4$  nanosheets and a higher stacking tendency in the  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  hybrid material. Doping with Mn appears to promote lateral sheet growth and partial agglutination of the nanosheets, due to interlayer interaction and surface charge distribution changes. As a consequence of induced changes in morphology brought about through doping with Mn, the electronic structure, charge carrier transport, and catalytic ability of the hybrid material are likely to be modified.

### 3.1.4. The effect of doping on elemental distribution

Energy-dispersive X-ray spectroscopy (EDX) analysis and elemental mapping were performed to assess the elemental composition and spatial distribution of constituent elements in the synthesized nanostructures. Fig. 4 presents the EDX spectra and elemental mapping images of  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$  and  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  composites. The elemental mapping confirms a homogeneous and uniform distribution of Bi, V, Ti, Mn, and O in the  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  composite, suggesting

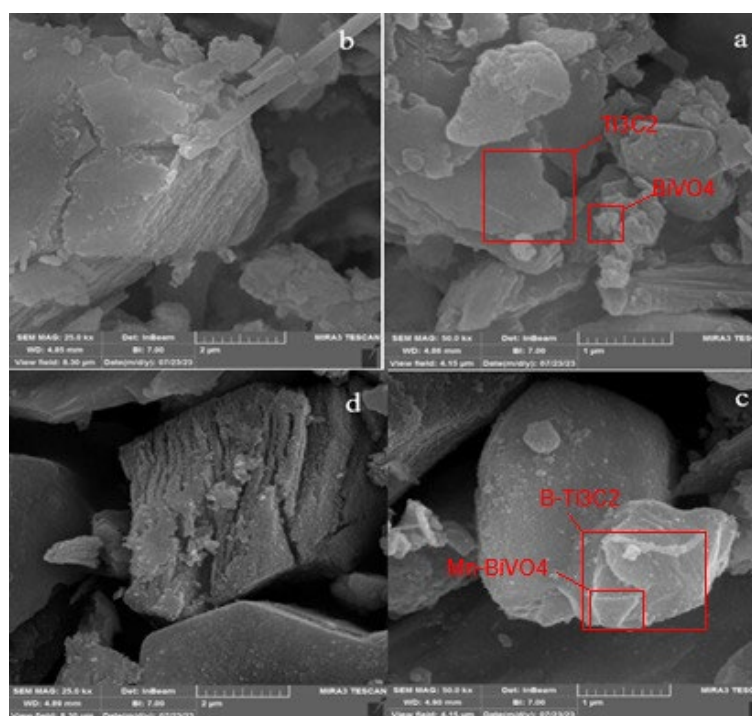


Fig. 3. FESEM images of (a,b)  $\text{BiVO}_4/\text{Ti}_3\text{C}_2$ , (c,d)  $\text{Mn-BiVO}_4/\text{B-Ti}_3\text{C}_2$  photocatalysts

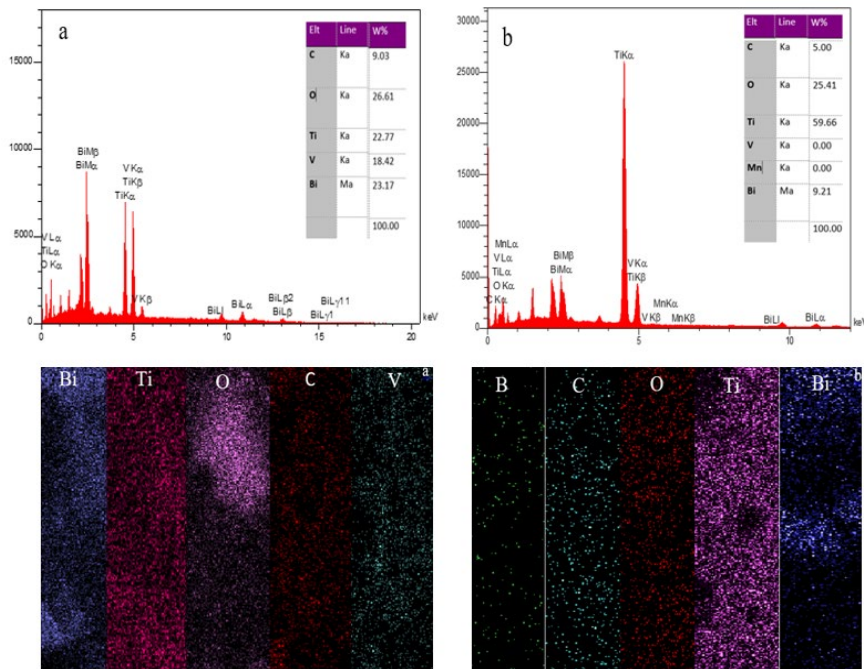


Fig. 4. EDX spectra and elemental distribution map of (a) BiVO<sub>4</sub>/Ti<sub>3</sub>C<sub>2</sub>, (b) Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub> photocatalysts

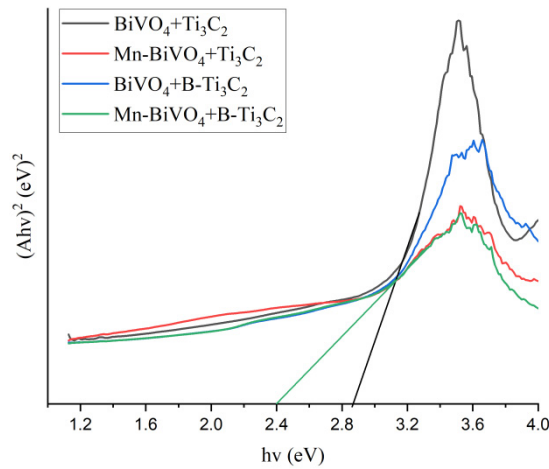


Fig. 5. Tauc plot of nanostructures

successful doping and coupling. This technique is not able to identify the boron element.

### 3.1.5. The effect of doping on band gap and optical properties

Ultraviolet-visible (UV-Vis) spectroscopy was utilized to evaluate the optical absorption characteristics. To determine the optical band gap (E<sub>g</sub>), Tauc's equation was employed:

$$(Ah\nu)^2 = B(h\nu - E_g)$$

where A represents absorbance, hν denotes photon energy, B is a proportionality constant, and E<sub>g</sub> is the band gap energy [32]. By extrapolating the linear portion of the Tauc plot (Fig. 5b), the direct optical band gap of Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub> was estimated to be 2.4 eV, which is lower than that of the undoped samples. This reduction in band gap energy can be attributed to the introduction of

Mn<sup>2+</sup> ions, which create localized electronic states within the band structure, facilitating enhanced photon absorption and improved charge carrier separation. These results confirm that Mn and B doping effectively tunes the electronic and optical properties of the BiVO<sub>4</sub>/Ti<sub>3</sub>C<sub>2</sub> system, making it a promising candidate for advanced photocatalytic and optoelectronic applications.

#### 4. Conclusion

This study demonstrates that Mn and B doping, combined with 2D/2D heterostructure formation, effectively modulates the structural, optical, and electronic properties of Ti<sub>3</sub>C<sub>2</sub> MXene and BiVO<sub>4</sub> nanosheets. The proposed Mn-BiVO<sub>4</sub>/B-Ti<sub>3</sub>C<sub>2</sub> hybrid displays enhanced light absorption, a narrowed band gap, and improved charge separation, indicating its great potential for photocatalytic hydrogen evolution. Future work efforts are intended to focus on the optimization of doping levels, elucidating charge transfer mechanisms, and evaluating long-term stability to advance practical applications in solar-driven water splitting and other photo-driven redox processes.

#### Conflicts of Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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